



## GENERATION OF NANO-/MICROSTRUCTURES OF CdS AND ZnO USING ELECTRODEPOSITION AND THEIR CHARACTERIZATION

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**Abstract:** The membranes used here as templates were of Makrofol KG foil (polycarbonate from Bayer AG), 10  $\mu\text{m}$  thick, having an average pore diameter  $\sim 2 \mu\text{m}$  with pore density  $2 \times 10^6 \text{ cm}^{-2}$ . This was prepared by irradiating the foil with  $\text{Si}^{8+}$ , energy ca. 100 MeV at  $90^\circ$  at the GPSC of the Pelletron at the IUAC, New Delhi, India, followed by chemical amplification of the damage trails by etching in 6 N NaOH, at  $50 \pm 2^\circ \text{C}$  for 35 min. SEM photograph displaying the top view of the synthesized microstructures. Earlier it has been reported that if the dimensions of the pores were small, it would suppress the fractal like cauliflower morphology, which is a usual feature in electrodeposited II–VI materials (Klein et al 1993). In the present case, a close view of the microstructures depicting the typical cauliflower morphology of the synthesized CdS microstructures is obtained. In order to fabricate CdS nanowires in AAMs, the cathode of the two-electrode deposition cell was closely covered with gold coated AAMs. The cathode was made highly smooth and the electrolyte and deposition conditions were maintained same as described for deposition through TEMs.

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**Keywords:** Templates, Nano/Microstructure, Electrodeposition, CdS and ZnO

### Introduction:

In recent years, there has been a rapid development in the field of II – VI group semiconductor nano-/microstructures, especially chalcogenides and oxides. Semiconductor nanostructures exhibit novel physical properties which make them suitable for potential applications in nanoscale electronics, optoelectronic devices, high –density magnetic memories, electrochemical and sensing devices (Cobden 2001; Ma et al 2003 ; Fasol 1996; Stafford et al 1997; Hoppleer and Zwrger 1999; Prinz 1998; Li et al 1997). The present study includes the fabrication of the II-VI group semiconductor compounds CdS and ZnO. CdS finds its applications as an important material for functional nanodevices such as light emitting diodes, sensors, solar cells (Kang and Kim 2006), thin film field effect transistors (Duan et al 2003), photoconductors, and optoelectronic devices (Woggon et al 1993). CdS is a well known n-type wide direct band gap II-VI semiconductor with  $E_g = 2.42 \text{ eV}$  at room temperature. So far, a variety of techniques such as physical vapor deposition (Wu and Tao 2002), chemical solution transport growth

(Zhan et al 2000, Zhang et al 2002), solvothermal synthesis, sulfurization (Fan et al 2002), microwave assisted growth (Yonghong et al 2004), thermal evaporation (Zhang et al 2006) and electrodeposition (Enculescu et al 2005, Kang and Kim 2006)) have been used to synthesize CdS nanowires. We report here the synthesis of CdS nano-/micro structures in TEMs and AAMs. The energy band gap ( $E_g$ ) value of CdS nanowires were estimated by absorption spectra using Tauc relation. Laser induced pulse excitation method was used to study the photoluminescence spectra.

Another material of choice in present research was ZnO which is a well known n-type wide direct band gap semiconductor with  $E_g = 3.37 \text{ eV}$  and large excitation energy (60 meV) at room temperature. ZnO is the most commonly studied material for functional nanodevices such as LEDs (Konenkamp et al 2004), gas sensors (Xu et al 2005, Wang et al 2006, Wan et al 2004), FETs (Fan et al 2004, 2006), piezoelectric-nanogenerators, gated diodes and resonators, field emitters (Liao et al 2005, Xu et al 2004, Zhao et al 2006), optical waveguides (Law et al 2004), dye-sensitized solar cells (Law et al

2005) and logic gates (Park et al 2005). Various growth mechanisms such as CVD (Bae et al 2004), direct thermal evaporation (Wang et al 2004), plasma-enhanced CVD (Liu et al 2004), metal – organic chemical vapor deposition MOCVD (Park et al 2002), electrochemical deposition (Leprince-Wang 2005), sol-gel deposition (Chen et al 2005), sonochemical and microwave-assisted (Hu et al 2004), surfactant assisted growth (Xu et al 2002) and hydrothermal processes (Liu and Zeng 2003) have been reported in literature to fabricate ZnO nanowires. We report here the fabrication of ZnO nanowires in AAMs using electrodeposition. The structural properties of the synthesized ZnO nanowires have been studied by SEM and XRD. Laser induced pulse excitation method was used to study the PL spectra.

Various types of electrodeposition cells have been used for synthesizing nano-/microstructures of metals (Penner and Martin 1987, Tierney and Martin 1989, Chakarvarti and Vetter 1991, Dobrev et al 1995). In general, a suitable cell offers a low ohmic voltage drop across the electrodes even if the cell is filled with poorly conducting electrolyte. The schematic diagram of the cell used in our laboratory is as shown in figure. The cell is a cylindrical structure made of perspex with a detachable cathode and anode assembly separated by a diaphragm of perspex sheet. For homogeneous growth of nano-/microstructures within the template, the anode is made preferably conical because for flat anodes, the material is deposited preferentially at the outer boarder areas of the cathode (Puipe and Leaman 1986). The area of anode is made much larger than the effective area of cathode in order to make anode overvoltage small compared to cathode overvoltage during deposition and anode is used at the same time as references electrode.

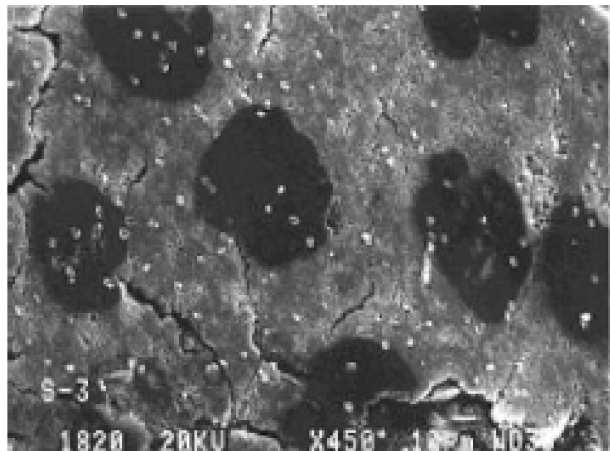
### **CdS microstructures in track etch templates**

The membranes used here as templates were of Makrofol KG foil (polycarbonate from Bayer AG),

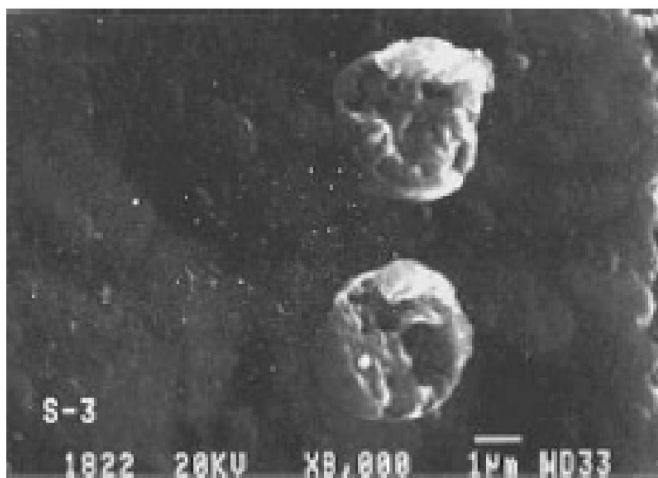
10  $\mu\text{m}$  thick, having an average pore diameter  $\sim 2 \mu\text{m}$  with pore density  $2 \times 10^6 \text{ cm}^{-2}$ . This was prepared by irradiating the foil with  $\text{Si}^{8+}$ , energy ca. 100 MeV at  $90^\circ$  at the GPSC of the Pelletron at the IUAC, New Delhi, India, followed by chemical amplification of the damage trails by etching in 6 N NaOH, at  $50 \pm 2^\circ \text{C}$  for 35 min. In order to produce see-through pores, optimum etch time and etch conditions are pre-set. For the fabrication of the CdS microstructures in the form of columnar ensembles, the cathode (Al tape as substrate having its base coated with conducting adhesive) of the specially designed electrodeposition cell, having provisions for temperature control and stirring at different speeds, was covered closely with the processed TEM and the electrolyte solution to be used in the bath was prepared using milli Q 10-M $\Omega$  water and ultrahigh purity reagents 0.002M  $3\text{CdSO}_4 \times 8\text{H}_2\text{O}$  (98%) + 0.1M  $\text{Na}_2\text{SO}_3$  (99.9%). The pH of the electrolyte was adjusted between 2-3 using dilute  $\text{H}_2\text{SO}_4$ . The galvanic replication was carried out for 50 min at 3.5 V at a room temperature of  $29 \pm 1^\circ \text{C}$  with an anode of pure Cd.

### **SEM Characterization**

Images were recorded on the photographic film at different magnifications revealing the finer details of the microstructures and the etched pores of the host template. Fig 1 shows SEM photograph displaying the top view of the synthesized microstructures. Earlier it has been reported that if the dimensions of the pores were small, it would suppress the fractal like cauliflower morphology, which is a usual feature in electrodeposited II–VI materials (Klein et al 1993). In the present case, a close view of the microstructures depicting the typical cauliflower morphology of the synthesized CdS microstructures is obtained and shown in Fig 2. This is expected because of the fact that the pore size here was not in the nanorange. It has been reported (Bandyopadhyay and Miller 2000) that bright field high-resolution transmission electron microscopy of the materials deposited in the pores show a crystalline structure and crystallinity can further be increased by thermal annealing.



**Fig. 1 Top view of the synthesized CdS microstructures**



**Fig. 2 Fractal-like cauliflower morphology of the synthesized CdS microstructures**

### **CdS Nanowires in anodic alumina membranes**

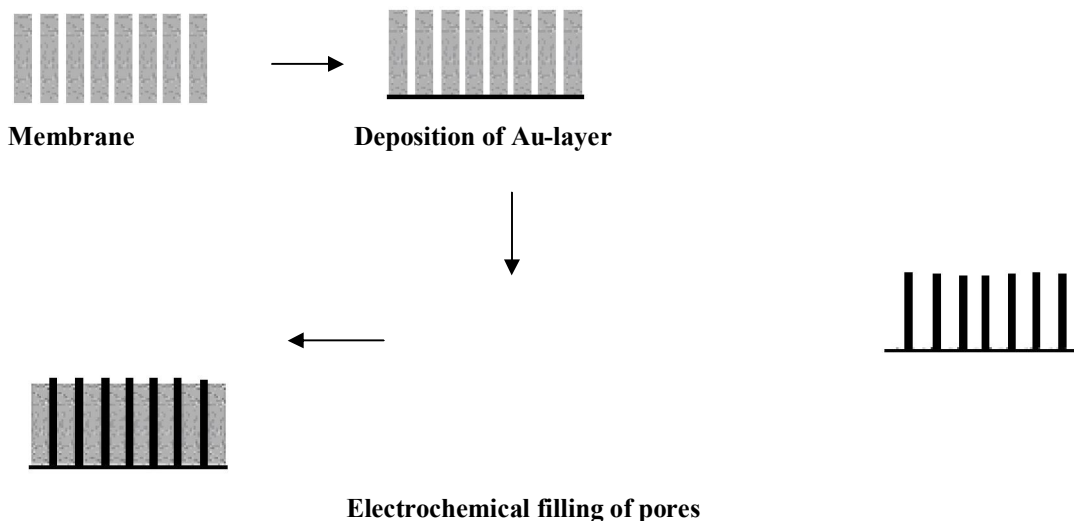
Commercially available AAMs having 20nm, 100nm pore diameters were used for electrodeposition of nanowires. In order to provide a conductive path for electrodeposition, a thin layer of gold ~100nm thick was deposited on the bottom surface of AAMs. In order to fabricate CdS nanowires in AAMs, the cathode of the two-electrode deposition cell was closely covered with gold coated AAMs. The cathode was made highly smooth and the electrolyte and deposition conditions were maintained same as described for deposition through

TEMs. The membrane supported electrodeposition process is shown in Fig 3.

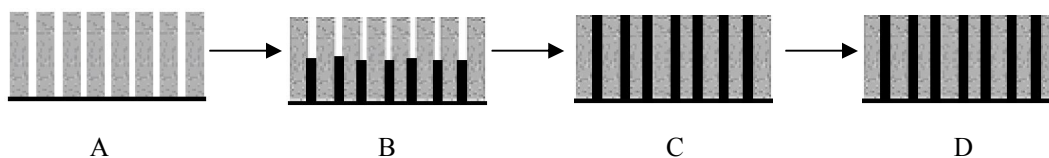
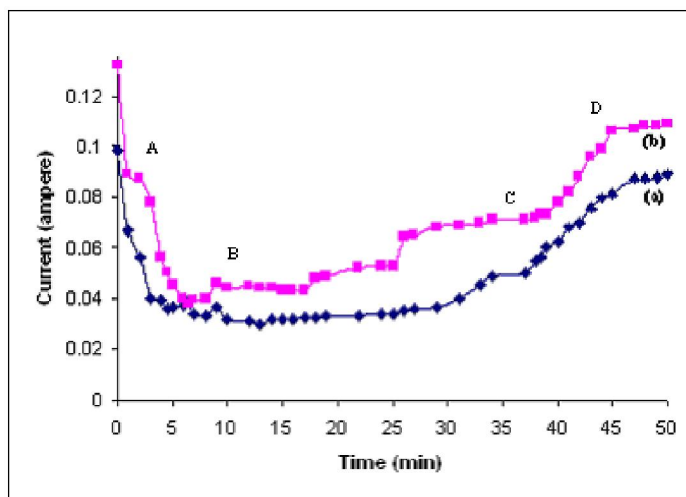
Fig. 4 shows the current vs time response for potentiostatic deposition of CdS nanowires of 100nm (a) and 200nm (b). The current vs time graph show that the deposition of nanowires is not a steady state process. After applying the potential, the current dropped suddenly due to creation of diffusion layer within the pores of template and afterwards the current gradually increases during the growth of nanowires in the template and attains a steady state when the filling reached the top of template. As soon

as caps start to cultivate on the top of structure, the current starts increasing slowly which increases

rapidly with the increase of surface area of the caps due to overdeposition.



**Fig. 3** Membrane supported electrodeposition process Arrays of nano-/micro structures and retrieved after dissolution of membrane.



**Fig. 4** Current vs time response for potentiostatic deposition of CdS nanowires of 100nm (a) and 200nm (b)

**Structural csharacterization of CdS nanowires**

After the deposition was over, the nanowires were released for further investigation, by dissolving

the AAMs in 1M NaOH solution for 1 h followed by rinsing with double distilled water. Figures 5 and 6 show the photographs of CdS nanowires viewed

under scanning electron microscope operated at 10kV.

#### Optical characterization of CdS nanowires

The fundamental optical property of CdS nanowires investigated here is the absorption spectra at various wavelengths using UV-visible spectrophotometer. Fig 7 and 8 show the absorption spectra of CdS nanowires of 100 and 200nm respectively. The energy band gap of CdS nanowires is determined from absorption spectra using Tauc relation:

$$\alpha hv = A (hv - E_g)^n,$$

Where  $\alpha$ ,  $hv$ ,  $E_g$ ,  $n$  and  $A$  have usual meanings as described earlier in Chapter-I.

The graphs between  $(\alpha hv)^2$  vs  $hv$  for CdS nanowires of 100 and 200nm are shown in figure 9 and 10 respectively. The presence of single slope in curve (fig. 9) indicates the direct and allowed transitions. The extrapolation of straight line to  $hv$  axis gives value of energy gap value of CdS. The slight increase in band gap of nanowires is observed as the diameter of nanowires varies from 200nm to 100nm.

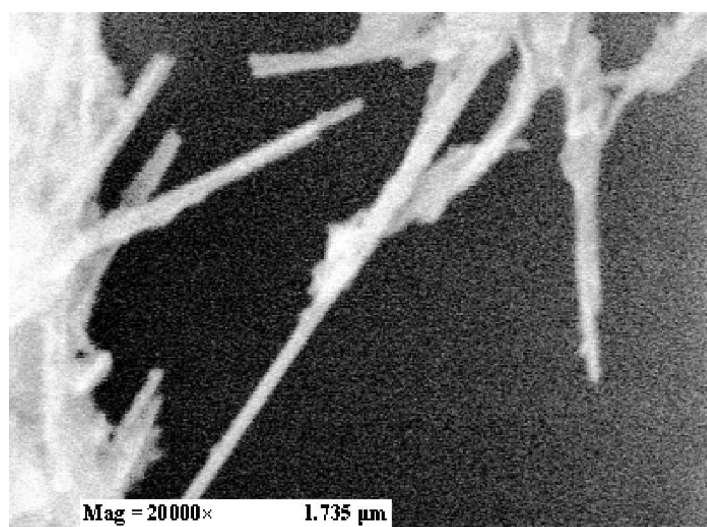


Fig. 5 SEM photographs of CdS nanowires (100nm)

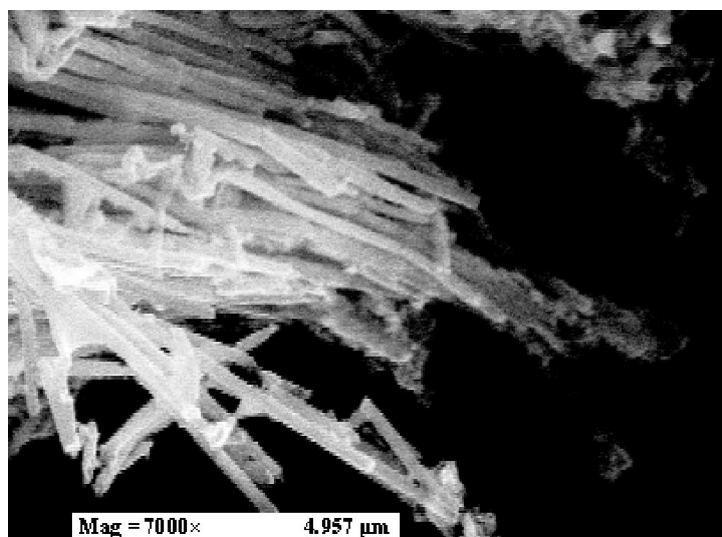


Fig.6 SEM photographs of CdS nanowires (200nm)

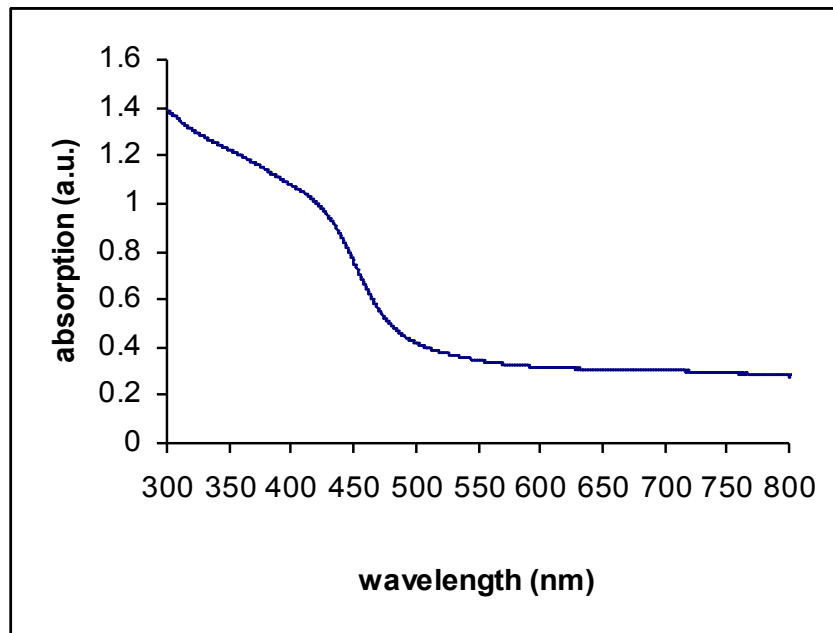


Fig. 7 Plot of absorption ( $\alpha$ ) versus wavelength ( $\lambda$ ) for CdS nanowires (100nm)

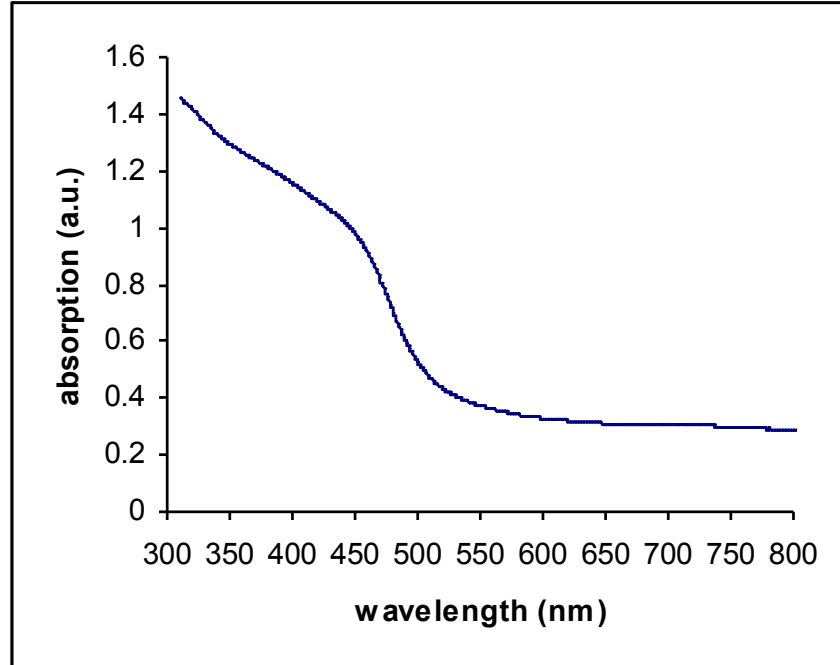


Fig. 8 Plot of absorption ( $\alpha$ ) versus wavelength ( $\lambda$ ) for CdS nanowires (200nm)

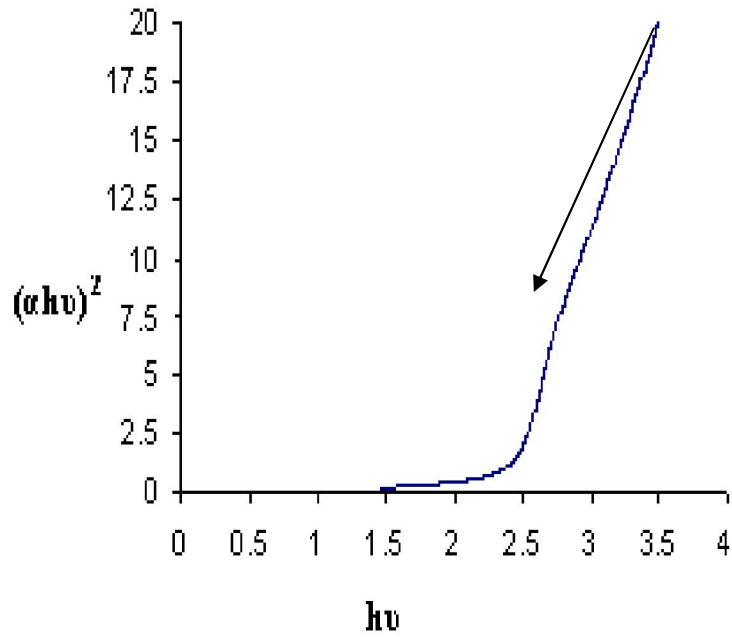


Fig. 9 Plot of  $(\alpha h\nu)^2$  versus  $h\nu$  (Tauc plot with dotted line is a theoretical fit) for CdS nanowires (200nm)

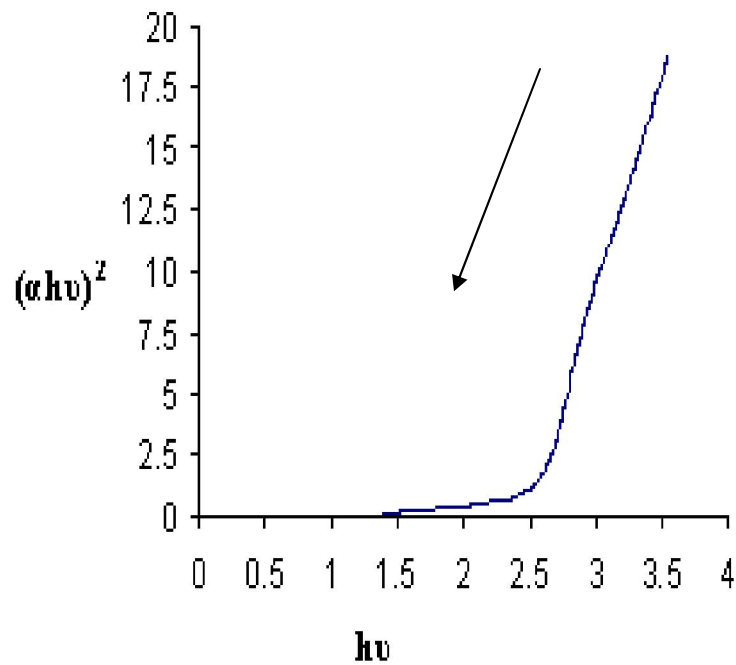
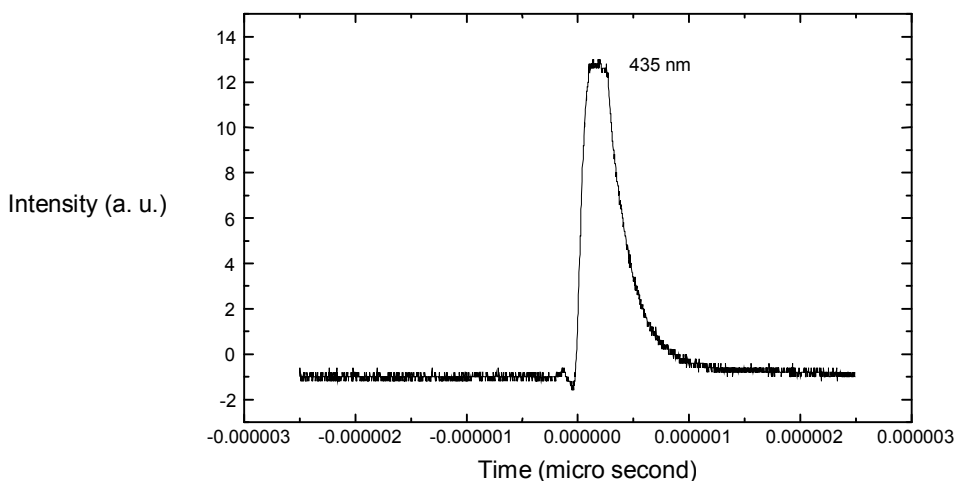


Fig. 10 Plot of  $(\alpha h\nu)^2$  versus  $h\nu$  (Tauc plot with dotted line is a theoretical fit) for CdS nanowires (100nm)

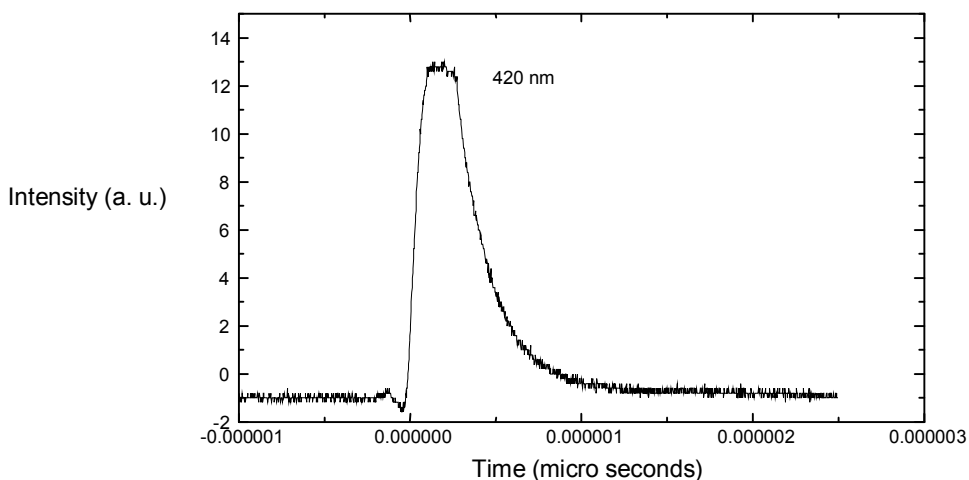
**Photoluminescence (PL) study of CdS nanowires**

Further, laser induced time resolved photoluminescence was used to study the photoluminescence spectra of 100 and 200nm CdS nanowires. The decay signals (Fig. 11 and 12) from the CdS nanowires were recorded for calculating emission wavelength of the nanowires. Photoluminescence investigation shows that the 200 nm CdS nanowires have an intensive and broad PL emission band peaked around 435 nm and with decrease in the size of CdS nanowires to 100 nm only

slight variation is observed in the photoluminescence wavelength i.e 420 nm which shows that the electrons movement is slightly confined due to the size reduction but as the size of these wires is more than the Bohr exciton radius of CdS (5 nm), the quantum confinement is not observed. The results suggest that localized defects and excess S atoms existing in the CdS nanowires are responsible for this blue luminescence as no dopant had been added to the host of CdS nanowires.



**Fig. 11 Decay curve of 200 nm CdS nanowires**



**Fig. 12- Decay curve of 100 nm CdS nanowires**



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